

## EFFECTS OF Sr-SUBSTITUTION ON THE MICROSTRUCTURE AND MAGNETIC BEHAVIOR OF M-TYPE HEXAGONAL FERRITES SYNTHESIS BY CO-PRECIPIATION METHOD

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In this work, the  $Ba_{1-x}Sr_xFe_{12}O_{19}$  hexaferrites ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) were synthesized by using co-precipitation method and sintered at a temperature of  $1100\text{ }^\circ\text{C}$  for 8h. The investigation of structural and magnetic properties was carried out for the samples using X-ray diffraction (XRD) and magnetometer also employed to measure the magnetic properties. XRD reveals that the structure of these nanoparticles is hexagonal phase and average crystallite sizes calculated from Debye Scherrer formula lies in the range of (27.03-74.4 nm). Lattice parameters were found to be increases linearly with Sr-concentrations. Grain size estimated from scanning electron microscopy (SEM) and found to be increase with increasing  $Sr^{+3}$  content. The elemental analysis was done by EDXS and it is close agreement with the expected composition from the stoichiometry of the reactant solutions. Magnetic measurements showed that coercivity and retentivity increases from (1.22-1.47 kOe) and (1.30 - 1.70 kGauss) that varies significantly with  $Sr^{+3}$  Content. The obtained results show that the investigated materials are potential candidates for magnetic recording media and high frequency applications.

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**Keywords:** Co-precipitation Method, M-Type hexaferrites, X-ray diffraction, B-H and J-H loops

### 1. Introduction

The M-type hexagonal ferrites ( $BaFe_{12}O_{19}$ ) are an interesting class of ferromagnetic oxides. In M type hexagonal structure  $Fe^{+3}$  are distributed in five distinct crystallographic sites, one tetrahedral ( $4f_1$ ), three octahedral ( $2a, 12k$  and  $4f_2$ ) and one trigonal bipyramidal ( $2b$ ) sites. In Ba-hexaferrites, along crystallographic c-axis spins of three sites  $12k$ ,  $2b$  and  $2a$  sites are aligned and parallel to each other when in the magnetically ordered state while spins of  $4f_2$  and  $4f_1$  sites point in the opposite direction. Understanding the effects of various diamagnetic and paramagnetic cations on the magnetic properties of  $Sr^{+2}$  substituted hexaferrites is one of the most important tasks associated with use of these nanomaterials in variety of modern technological applications [1-2]. Due to high resistivity, power and eddy current loss and decreased remarkably and consequently make the synthesized ferrite materials more suitable in high frequency device fabrications. Several techniques have been developed and effectively employed for synthesis of hexaferrites such as ceramic method [3-4] solution combustion technique [5] sol-gel auto combustion process [6-7] ball milling [8] and co-precipitation technique [9-10]. Co-precipitation is one of the simplest techniques for the preparation of hexaferrite nanomaterials. In current study, we have reported the synthesis of  $Ba_{1-x}Sr_xFe_{12}O_{19}$  with ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) nano-hexaferrites, prepared by co-precipitation technique. In addition to that, the influence of  $Sr^{+2}$

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concentrations on the structural and magnetic properties of  $\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19}$  nanomaterials were investigated.

## 2. Materials and Methods

### 2.1. Samples preparation and equipments

$\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19}$  hexaferrite samples with ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) were synthesized by chemical co-precipitation technique [11-12]. The reagents  $\text{Ba}(\text{NO}_3)_2$ ,  $\text{Sr}(\text{NO}_3)_2$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and  $\text{NaOH}$  are of analytical grade, de-ionized water was used to make the precursor. Nitrates and salts of metals with stoichiometric composition of  $(\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19})$  were mixed in 300 mL de-ionized water. The precipitating agent was mixed drop wise during stirring until the co-precipitation occurs. When the precipitates were formed, the solution was washed thoroughly with ethanol. The precipitates were then filtered and dried at  $100^\circ\text{C}$  for 10 h in oven. By using a mortar and pestle, the dried powder was grinded homogenously by grinding the constituents for 1 h. The pellets were formed in the hydraulic press (Paul-Otto Weber) at a pressure of 30 KN for about 3 min. The pellets were pre-sintered in a digital electric furnace for 6 h at temperature  $1000^\circ\text{C}$  in an ambient atmosphere. After each heat treatment the samples were extinguished in air under to have equilibrium position of the cations. The X-ray diffraction (XRD) patterns were obtained at room temperature using powder samples by using Xpert Pro PANalytical diffractometer with  $\text{Cu-K}\alpha$  radiation ( $\lambda = 1.54056 \text{ \AA}$ ) at 40 kV and 30 mA. Intensity data was collected by the step counting method (with a scanning speed  $0.05^\circ/\text{s}$ ) in the  $2\theta$  range from  $20^\circ - 70^\circ$ . The surface morphology and microstructure of the samples were studied by JSM-6490 JEOL scanning electron microscope (SEM). The elemental composition was determined by energy dispersive peak of the representative sample using energy dispersive X-ray spectroscopy (EDXS, Model JFC-1500 JEOL). Magnetometer was used to measure the magnetic properties with a maximum applied magnetic field.

### 2.2. Calculations

The structural parameters such as lattice parameters ('a' and 'c'), Unit cell Volume ( $V_{\text{cell}}$ ), Crystallite size (D) and X-ray density of  $\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19}$  M-type hexaferrites were calculated from XRD data using the following formulae [13-14].

$$\sin^2\theta = A(h^2 + hk + k^2) + C l^2 \quad (1)$$

Where  $A = \frac{\lambda^2}{3a^2}$  and  $C = \frac{\lambda^2}{4c^2}$

$$V_{\text{cell}} = a^2 c \sin 120^\circ \quad (2)$$

$$D = \frac{k \lambda}{B_{(hkl)} \cos \theta} \quad (3)$$

$$\rho_{\text{X-ray}} = \frac{Z M}{N_A V} \quad (4)$$

Where 'a' and 'c' are lattice constant, (hkl) are Miller indices, V is the unit cell volume, Z represents 2 molecules per unit cell of the hexagonal structure,  $N_A$  is the Avogadro's number ( $6.02 \times 10^{23} \text{ g/mol}$ ) and M is the molecular weight of the sample. Scherer's formula ascribed by equation 3 was used to calculate the crystallite size (D) nm, where, K is the shape factor,  $\lambda$  is the X-ray wavelength (1.54 Å) and  $\theta$  is the Bragg's diffraction angle.

### 3. Results and discussion

#### 3.1. Structural analysis

Fig.1 shows the X-ray diffraction patterns of M-type hexaferrite  $Ba_{1-x}Sr_xFe_{12}O_{19}$  ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ). The XRD patterns were indexed and observed d-values of all the peaks were compared with JCPDS Card (00-043-0002). The diffraction peaks at  $2\theta$  values are related to the diffraction planes (105), (107), (108), (109), (118), (302), (214), (217), (2013), (228) and (402). The analysis of XRD patterns indicates that the observed peak corresponds to the M-type hexagonal phase and single diffraction peak of  $\alpha-Fe_2O_3$  is also observed at  $2\theta = 33.2$ . Different parameters such as crystallite size ( $D$ ), lattice constants ('a' and 'c'), unit cell volume ( $V_{cell}$ ) and X-ray density ( $\rho_x$ ) calculated from the XRD data and their values were listed in (Table 1.) The lattice parameters ('a' and 'c') and unit cell volume ( $V_{cell}$ ) were calculated by applying the following equations 1 and 2 [15]:

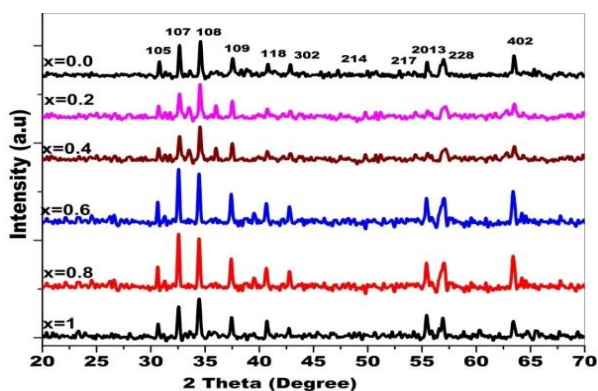


Fig.1. XRD patterns of  $Ba_{1-x}Sr_xFe_{12}O_{19}$  M-type hexaferrite powders ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) ferrites.

The lattice parameters  $a = (5.78 - 5.83\text{\AA})$  and  $c = (23.12 - 23.27\text{\AA})$  increase steadily with increasing the substituted amount of  $Sr^{2+}$  ions into the hexagonal lattice as shown in Fig.2. The observed variation in the lattice parameters can be explained on the basis of the relative ionic radii of  $Sr^{2+} = (1.32\text{\AA})$  and  $Ba^{2+} = (1.35\text{\AA})$  ions, in this case ionic radii of  $Ba^{2+}$  is larger than ionic radii  $Sr^{2+}$ , This may result in an increase in the lattice constants. The unit cell volume also increases from  $(668.70 - 685.41\text{\AA}^3)$  regularly with increasing strontium content (Table 1) which may be due to increase in both the lattice parameters 'a' and 'c'. The obtained values of X-ray density listed in Table 1. The average crystallite size was calculated from the width of the most intense peak and was found to be in the range of  $(27.03 - 74.4\text{ nm})$  (Table 1).

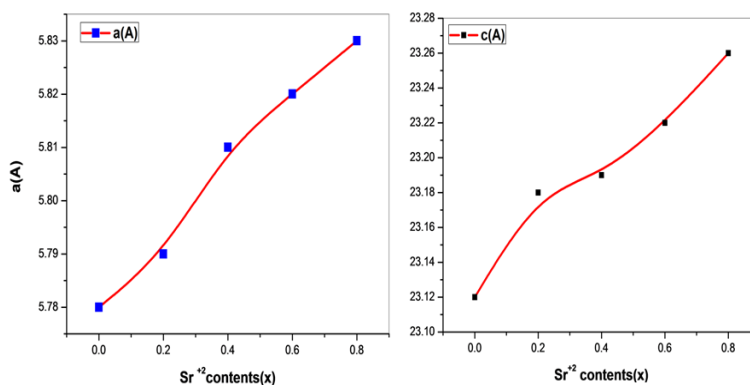


Fig.2. Variation of lattice constant 'a' and 'c' for  $Ba_{1-x}Sr_xFe_{12}O_{19}$  ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) ferrites on various  $Sr^{2+}$  contents

The requirement of the grain size for high density recording media is less than 50 nm to obtain the suitable signal to noise ratio [16]. The trend of the crystallite size varies by substitution strontium contents is revealed in the Fig.3

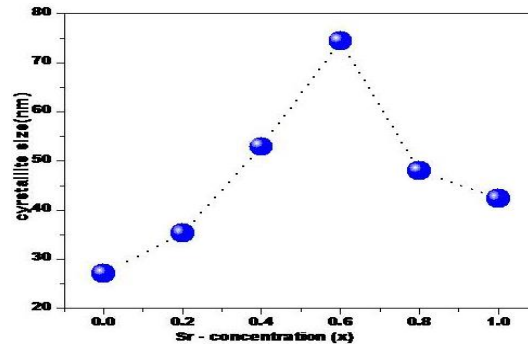


Fig.3. Crystallite size as a function of  $Sr^{+2}$  contents for  $Ba_{1-x}Sr_xFe_{12}O_{19}$  ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) ferrites on various  $Sr^{+2}$  contents.

Table.1. Lattice parameters ('a' and 'c'), Unit cell Volume (V cell), Crystallite size (D), and X-ray density of  $Ba_{1-x}Sr_xFe_{12}O_{19}$  hexagonal ferrites

Composition	Lattice Parameter (a) Å	Lattice Parameter (c) Å	Unit Cell Volume (Å <sup>3</sup> )	Crystallite Size (D nm)	X-ray density (g/cm <sup>3</sup> )
$BaFe_{12}O_{19}$	5.78	23.12	668.90	27.03	3.73
$Ba_{0.8}Sr_{0.2}Fe_{12}O_{19}$	5.79	23.15	672.08	35.3	3.72
$Ba_{0.6}Sr_{0.4}Fe_{12}O_{19}$	5.80	23.18	675.75	52.9	3.74
$Ba_{0.4}Sr_{0.6}Fe_{12}O_{19}$	5.81	23.24	679.84	74.4	3.73
$Ba_{0.2}Sr_{0.8}Fe_{12}O_{19}$	5.82	23.25	682.00	48.0	3.73
$SrFe_{12}O_{19}$	5.83	23.27	685.41	42.3	3.74

### 3.2. Scanning Electron Microscopy (SEM)

The SEM micrographs of the synthesized samples  $Ba_{1-x}Sr_xFe_{12}O_{19}$  as a function of  $Sr^{+2}$  concentrations(x) are shown in Fig.4 (a, b, c, and d). The majorities of grains are found to be relatively well-defined hexagonal platelets and are homogeneously distributed.

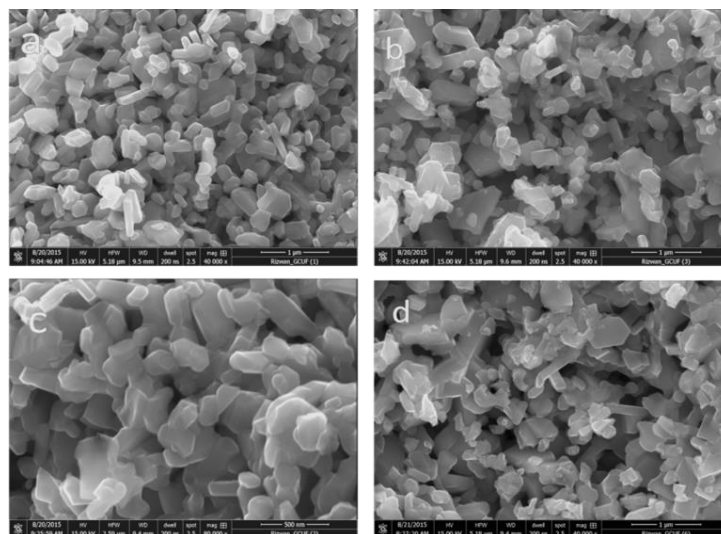


Fig.4 (a, b, c, and d) SEM images of  $Ba_{1-x}Sr_xFe_{12}O_{19}$  M-type hexa ferrite

The hexagonal platelet-like shape particles with smaller aspect ratio in morphology are most suitable for applications in microwave absorption materials. Xiaogu et al. [17] and Wanga et al. [18] also reported that samples of hexagonal flake shape are beneficial for microwave absorption. In the present study, the average crystallite size as calculated from XRD by Scherer formula is smaller than the average grain size derived from SEM measurements. It means that every grain is formed by aggregation of a large number of crystallites. The shape of the materials is very important for various applications in different fields.

### 3.3. Energy dispersive X-ray spectroscopy (EDXS)

Figure 5 (a, b, c and d) shows the selected M-type hexaferrites composition of the synthesized samples was determined by energy dispersive X-ray spectroscopy (EDXS) analysis. It is evident from the analysis that the stoichiometric amount of  $\text{Sr}^{2+}$  increases while that  $\text{Ba}^{2+}$  content decreases indicating that strontium is being substituted at barium site. The labeled peaks in the figure clearly indicate the presence of Ba, Sr, Fe and O.

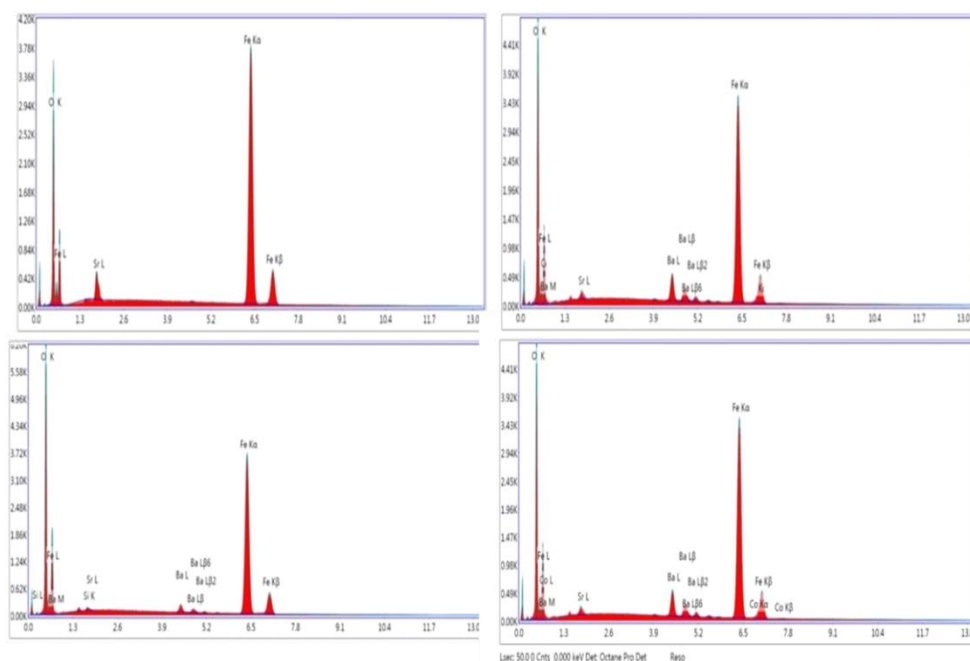


Fig. 5. EDX spectrums of  $\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19}$  hexaferrite

### 3.4. Magnetic Properties

The field dependent B–H and J–H loops  $\text{Ba}_{1-x}\text{Sr}_x\text{Fe}_{12}\text{O}_{19}$  ( $x = 0.0, 0.2, 0.4, 0.6, 0.8$  and  $1.0$ ) are shown in Fig. 6 (a, b, c, d and e). The upper loop represents the B–H curve and the lower loop indicates J–H curve. These hysteresis loops (B–H and J–H curves) play a very important role in comparing the magnetic properties like coercivity, loop area and retentivity by using magnetometer at room temperature. From B–H loop the magnetic parameters such as energy product ( $\text{BH}_{\text{max}}$ ), coercivity ( $\text{H}_{\text{cB}}$ ) and retentivity ( $\text{Br}$ ) were measured. These magnetic parameters are strongly affected by the occupation of substituted strontium cations at different sites in the structure of hexaferrite. The values of retentivity lies in the range from  $(1,307 \times 10^3 - 1.708 \times 10^3 \text{ Gauss})$  as listed in (Table 2). The retentivity is found to be enhance from  $(1.307 \times 10^3 - 1.708 \times 10^3 \text{ Gauss})$  with increase in  $\text{Sr}^{+2}$  contents ( $x$ ). The value of retentivity is maximum for  $\text{SrFe}_{12}\text{O}_{19}$  ( $x = 1.0$ ) i.e.  $1.708 \times 10^3 \text{ Gauss}$ .  $\text{BH}_{\text{max}}$  (maximum energy product) lies in the range  $(0.399 - 0.649 \text{ MgOe})$  as listed in (Table 2). The  $\text{BH}_{\text{max}}$  for unsubstituted was  $0.399 \text{ MGaOe}$  while it is highest i.e.  $0.649 \text{ MGaOe}$  for the composition  $\text{SrFe}_{12}\text{O}_{19}$  ( $x = 1.0$ ).  $\text{H}_{\text{cB}}$  is found to be  $1.233 \text{ kOe}$  for the composition  $\text{BaFe}_{12}\text{O}_{19}$  ( $x = 0.0$ ) it increases gradually from  $(1.223 - 1.473 \text{ kOe})$  with increase in  $\text{Sr}^{+2}$  concentration ( $x$ ). The enhancement in  $\text{H}_{\text{cB}}$  was mainly due to the increase  $\text{Sr}^{+2}$  concentration and due to substitution of  $\text{Sr}^{+2}$  ions with  $\text{Ba}^{2+}$  ions. If nanomaterials have

coercively values above 1200 Oe then these can be very useful in a new developing technology of magnetic recording as perpendicular recording media [19].

Table.2. Values of magnetic parameters retentivity ( $B_r$ ), coercivity ( $H_cB$ ),  $BH_{max}$  and  $H_cJ$  for different values of  $x$  in  $Ba_{1-x}Sr_xFe_{12}O_{19}$  M-type hexa ferrites.

X	$B_r \times 10^3$ ( Gauss)	$H_cB$ (kOe)	$BH_{max} \times 10^{-1}$ (MGaOe)	$H_cJ$ (kOe)
0.0	1.307	1.223	3.990	5.044
0.2	1.346	1.261	4.230	5.168
0.4	1.409	1.309	5.909	5.175
0.6	1.529	1.460	5.670	5.182
0.8	1.552	1.457	5.730	5.135
1.0	1.708	1.473	6.490	2.854

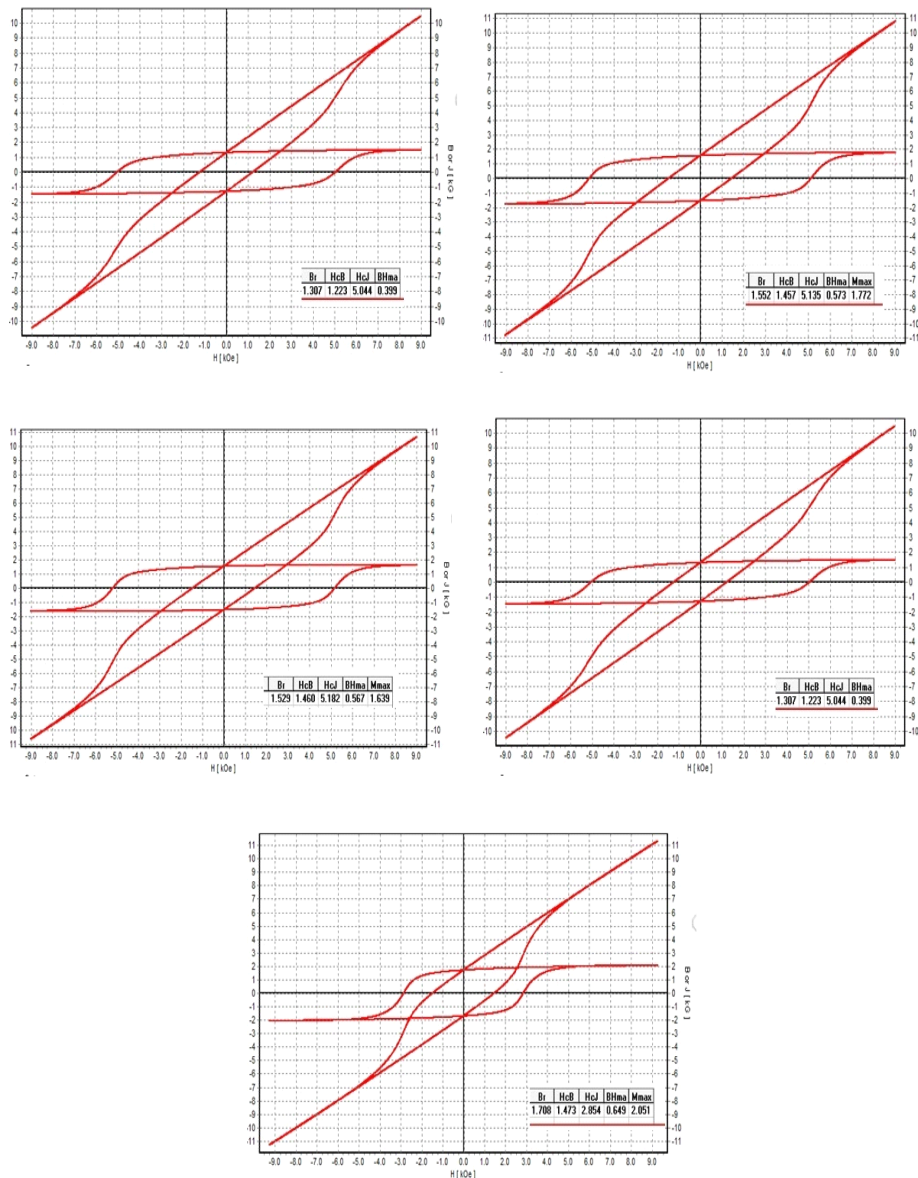


Fig.6. M-H loops of  $Ba_{1-x}Sr_xFe_{12}O_{19}$  hexaferrites ( $x=0.0, 0.2, 0.4, 0.6$  and  $1.0$ )



Nanomaterials synthesized in the present work have coercivity  $H_cB$  values in the range (1.307 -1.708 kOe). This implies that these nanomaterials may have applications in high density magnetic recording Media. In current study, all the synthesized nanomaterials have  $H_cB > B_r/2$  (for any fixed value of  $Sr^{+2}$  ions concentration) so for high frequency applications these nanomaterials can be very beneficial as reported in the literature [20]. The sample  $Sr Fe_{12}O_{19}$  ( $x = 1.0$ ) was found to exhibit best properties among all the synthesized samples because the magnetic parameters approached highest values as shown in (Table2). In addition to that, higher values of retentivity ( $B_r$ ) and coercivity ( $H_cB$ ) rendered these hexaferrite nanomaterials most beneficial for permanent magnets and for high density recording media.

#### 4. Conclusions

Co-precipitation method effectively produced  $Sr^{+2}$  substituted hexaferrites nanocrystalline. The effect of  $Sr^{+3}$  content on the structural parameters have been studied by X-ray diffraction and found to be in hexagonal phase. The physical parameters of the material revealed that crystallite size were found in range (27.03-74.4 nm) while lattice parameters increase with the increasing of  $Sr^{+3}$  concentrations. Magnetic measurements show that the highest substituted  $Sr^{+2}$  ( $x = 1.0$ ) is found to the best parameters such as energy product, retentivity ( $B_r$ ) and coercivity ( $H_cB$ ). These samples are useful in the magnetic recording applications, because recording media requires a high saturation magnetization and a moderately high coercivity

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