Synthesis and optical properties of the glassy compound As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}

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The amorphous calcogenide semiconductor $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ was synthesized and thin films based on it were obtained. XRD and optical images investigations showed the amorphous and homogeneous nature of the samples. The optical transmission spectrum in the UV-Vis range of $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ thin films shows good transparency in the spectral range 0.7-3.5 µm with a single absorption threshold at 2.05 eV and with refractive index in infrared 2.3. The irradiation of films with light leads to a parallel shift of the transmission spectrum to the IR range. The materials with $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ composition have high optical transparency have high optical transparency that make they promises for applications in holographic memory devices, optical amplitude and phase recorder, optical processing units and others.

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1. Introduction

Multicomponent glasses based on *As-S-Sb-Te* are intensively studied due to the ability to form new optical properties such as increased structural stability, flexibility, elasticity, etc. [1]. They are beginning to be widely applied in optoelectronics as infrared optical elements, as materials for image creation and holographic information storage, acousto-optical elements, optical memory switches [2,3], X-ray imaging [4] etc. These materials are less affected by temperature because they have a low level of phonons and have heavy anions. Sulphur, selenium and tellurium are the main components of their compositions. Multicomponent glasses are also being studied for applications in passive and active devices such as fiber laser amplifiers and nonlinear components [1]. Chalcogenide materials exhibit unique properties such as photodarkening, huge photoexpansion, and photofluidity when irradiated by appropriate light. Also, glasses are prone to non-linear optical properties.

This paper presents the extension of our previous experimental studies [5,6] regarding new chalcogenide materials in quaternal As-S-Sb-Te system. As shown in our early paper [5] chalcogenide materials, which have the composition similar to this one, namely $As_{11.2}S_{48.0}Sb_{28.8}Te_{12.0}$ and $As_{20.8}S_{48.0}Sb_{19.2}Te_{12.0}$ demonstrate polycrystalline structure. Thereby, the tusk of obtaining complex chalcogenide materials with high homogeneity remains a topical issue. In order to combine the advantages of short distance and medium-intermediate distance and with the increase of flexibility and elasticity in a single composition, in this paper an optimal quaternary vitreous compound with composition $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ was obtained. Optical properties of thin films obtained by thermal vacuum evaporation were studied.

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2. Experimental part and discussion

2.1. Synthesis of glassy material

All chemicals were purchased from Aldrich Chemical Co. The glassy chalcogenide semiconductor $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ was synthesized using the elements As, S, Sb, Te with a purity of (99.999%), in quartz ampoules. The precursors were weighed, loaded into the ampoule which was then evacuated, vacuum sealed (10^{-5} mm Hg) and placed in the oven. The furnace temperature was slowly raised at a rate of 1°C/min up to 920 °C. The maximum temperature of the liquid melt mixture was maintained for 12 hours along with the rotation of the furnace around the perpendicular axis to obtain a homogeneous mass. Subsequently, melt quenching was applied. The ampoule was suddenly cooled by removing it from the oven to room temperature. Parts of the synthesized ingot were ground into powder with grain sizes of about 100 nm for microscopic measurements and thin films preparation. The measurements of the elemental thin film's composition in different places of the sample showed the same composition, which further confirms the elemental homogeneity of the sample.

	Elements:	S	As	Sb	Te
·	Calculated, %	13.93	25.55	49.23	11.29
	Average measured, %	12.73	25.05	50.06	11.72

Table 1. EDX statistics of the elemental composition of the composite $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$.

2.2. Preparation of thin films

The thin films were thermally deposited by vacuum evaporation (10⁻⁵ mm Hg) using a "quasi-closed" complex molybdenum evaporator. A special cover with many holes is provided in the construction of the vaporizer to ensure a uniform flow on the substrate so that a uniform film thickness can be obtained over an area of 100 cm² [7, 8]. For this purpose, the VUP-4 vacuum plant was equipped with the recording and maintenance of the evaporator temperature.

Temperature control was carried out automatically with the VRT-2 temperature regulator. The films were obtained on optical glass substrates. The evaporation temperature of the $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ material was constant during the evaporation which ensured the maintenance of a constant condensation rate during the evaporation period. The thickness of the deposited layers was determined by the evaporation time. The thickness homogeneity was regulated by diaphragms mounted above the evaporator. The uniform thickness distribution over the whole layer was also ensured by selection of distance between the substrate and the evaporator. The distance between the evaporator and the substrate was equal to 21 cm. Transparent amorphous films with thicknesses of $125 \div 1000$ nm were obtained. The films thickness was measured using the MII-4 interferometric microscope. The structure of bulk material and thin films investigated by the XRD method shows the absence of any sharp line diffraction, indicating that the nature of the material is amorphous. The elemental analysis studied by EDX [9] and shown in Table 1 confirmed the composition of the material As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}. Phillips [10], Mott [11] and Flank et al. [12] showed that the coordination number of atoms with covalent bonds in glass is determined by the 8-N rule, where N is the number of electrons in the outer electronic shell [21]. The 8-N rule suggests that the number of nearest neighbor atoms for (S, Te) and (As, Sb) are two and three respectively. As calculation shows, the nearest neighbor atoms number for chosen $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ composition of material is similar to As2S3 number of neighbor atoms.

2.3. Optical transmission

UV-VIS transmission measurements were performed at room temperature on samples of $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ thin films, using a Perkin Elmer UV/VIS/IR–960 spectrometer.

The system uses a grid PMT with a Peltier-controlled PbS detector to perform highperformance tests in the spectral range up to 3300 nm. UV-VIS resolution reaches 0.05 nm, while NIR resolution reaches up to 0.2 nm. Absorption/transmission spectra for all samples were recorded in the 400–3300 nm range.

The optical transmission spectrum $T(\lambda)$ of the thin layer of $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}(T(\lambda))$ in the range 500 – 3500 nm is shown in Fig. 1.



Fig. 1. The transmission spectrum of the $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ layer.

It is formed by a steep threshold in the range 500 - 550 nm followed by the range of 550-3500 nm where T(λ) is oscillatory. From the spectrum of transmission in the transparency region the absorption coefficient α , the refractive index *n* and the layer thickness *d* were calculated using the method of Swanepoel [13]. The interference effects occur due to multiple reflections between the interfaces of a thin film with air and substrate, when the thickness of the film is close to the coherence length of the light. This effect produces an interference pattern in the transmission spectrum. At wavelengths longer than the bandgap and neglecting the absorption and scattering of light in the IR range, the maximum transmission values correspond to the transmittance of the substrate alone, while the minimum transmission occurs when the reflections from the two surfaces of the film interfere in the opposite phase.

The refractive index of the chalcogenide layer (n) (Fig. 2) was calculated from the transmission data according to the formula [9]:

$$\mathbf{n} = [\mathbf{N} + (\mathbf{N}^2 - \mathbf{n}_s^2)^{1/2}]^{1/2}$$
(1)

$$N = 2n_s (T_M - T_m)/T_M T_{m+} (n_s^2 + 1)^{1/2}$$
(2)

where T_M and T_m are the maximum and minimum transmission values, and n_s is the refractive index of the substrate (for glass $n_s = 1.5$).

Film thickness was determined by examining the distance between adjacent maxima and minima using the following equation: $d = k\lambda_k$ where d is the film thickness, $\lambda_k = (\lambda_M + \lambda_m)/2$ is the wavelength and k is the order number. Maxima occurs at integer values of the order number, and minima occur at half integers [9].



Fig. 2. Refractive index versus wavelength for $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ amorphous compound.

2.4. Optical absorption

The absorption coefficient (Fig. 3) was determined according to the formula α =-ln(T(λ))/d, where T(λ) is the transmission coefficient, and d is the layer thickness. The optical absorption spectrum is the important tool for studying the band gap of the material. The optical absorption edge (the value of the forbidden band E_g) was analysed by the following relation [14]:

$$(\alpha hv) = A(hv - E_g)^m \tag{3}$$

where A is the fringe width parameter characterizing the film quality.

Optical energy gap of the material is calculated from the tangent of the linear part of this relationship. Calculations made from the graph give for A the value of 700 eV^{-1/2}cm^{-1/2}. The index *m* is determined by type of optical transition in material. As is well known, the parameter *m* has the value 1/2 for direct transition allowed and two for indirect transition allowed.



Fig. 3. Processed absorption spectrum for films of the composition $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$.

The method of determination the type of transition involves a graphical representation of the relations $(\alpha h\nu)^2$ or $(\alpha h\nu)^{1/2}$ versus hv [15, 16]. The straight line was obtained for the dependence $(\alpha h\nu)^{1/2} = f(h\nu)$ as shown in Fig. 3, which means that indirect transitions are allowed.

The bandgap (optical activation edge) E_g was determined by intersection of extrapolate linear dependence of the graph $(\alpha h\nu)^{1/2} = A(h\nu \cdot Eg)$ with the hv axis. The bandgap of 2.05 ± 0.1 eV for this compound was determined from the graph.

2.5. The sensitivity to the irradiation with violet light

The glassy film of composition As0.63S2.70Sb1.37Te0.30 was exposed to the violet light illumination of the 30 W lamp for 15 min. After that the transmission spectrum was measured. As is shown in Fig. 4 photoinduced parallel shift of the $T(\lambda)$ spectrum towards the infrared range by about 50 nm is observed. These induced effects can be used to form various optical components including waveguides and surface gratings, homo- and heteronuclear bonds that produce the amorphous density of state defects. The doubly covalently bonded chalcogen atom possesses a lone pair of electrons, which does not bond completely in the glass. These bonds are broken under UV illumination and the its number are change after relaxation.

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Fig. 4. Transmission sensitivity (1) without irradiation and (2) upon irradiation with violet light $(\lambda = 405 \text{ nm})$ for 15 min.

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3. Conclusions

The amorphous glassy semiconductor $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ was synthesized and thin films layers were obtained. The composition of thin films practically reproduces composition of bulk. The amorphous material possesses both high homogeneity and high optical transparency. Photoinduced changes in transmission spectrum under UV illumination were investigated. The dispersion of the refractive index was obtained for a wide spectral range. From the absorption spectrum interpreted as indirect transitions the 2.05 eV value of optical bandgap was obtained. The thin films obtained from the synthesized material possessed large photoinduced changes of transmission under UV irradiation. Due to that, the $As_{0.63}S_{2.70}Sb_{1.37}Te_{0.30}$ amorphous compound can be proposed for application in memory devices and registration media.

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