

Original Research

Effects of Dopant Concentration and Temperature on Upconversion Photoluminescence of Textured $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$: Yb^{3+} , Ho^{3+} Ceramics

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Abstract: The upconversion photoluminescence properties of $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ ceramics co-doped with Yb^{3+} , Ho^{3+} ions were investigated, aiming at their potential application in temperature sensing. X-ray diffraction confirmed the successful synthesis and pronounced textured characteristics of the ceramics. Under 980 nm laser excitation, the $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ($x = 0.01, 0.02, 0.03, 0.04$) textured ceramics exhibited three distinct emission bands at 549 nm, 660 nm and 759 nm, corresponding to the transitions $^5\text{F}_4 \rightarrow ^5\text{I}_8$, $^5\text{F}_5 \rightarrow ^5\text{I}_8$ and $^5\text{F}_5 \rightarrow ^5\text{I}_6$ of Ho^{3+} ions. Emission intensities increased with Ho^{3+} concentration, suggesting the absence of concentration quenching. Power-dependent studies confirmed two-photon processes as the origin of three emission bands. Temperature-dependent measurements revealed monotonic thermal quenching, with $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ showing the strongest decrease in fluorescence intensity. Luminescence analysis showed near-infrared emission at 759 nm delivered the highest sensitivity, achieving maximum absolute sensitivity of 0.0072 K^{-1} and relative sensitivity of $1.71\% \text{ K}^{-1}$. These results demonstrate that Yb^{3+} , Ho^{3+} co-doped $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ textured ceramics are promising candidates for luminescence-based temperature sensing applications.

Keywords: upconversion; Ho^{3+} ; temperature sensing; $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$; textured ceramics

1. Introduction

Luminescent materials doped with trivalent lanthanide rare-earth ions have attracted significant attention due to their distinctive spectroscopic properties, which include upconversion and downshifting spectral conversion [1–5]. Upconversion is a nonlinear optical process whereby two or more photons are absorbed, leading to the emission of a single photon with higher energy [6]. In this process, the incident long-wavelength radiation is converted to the shorter-wavelength, a feature characteristic of the anti-Stokes process. The downshifting process represents another optical phenomena in which a single high-energy photon is absorbed and subsequently re-emitted as a photon of lower energy [7]. In the downshifting process, the short-wavelength radiation is converted into longer-wavelength emission, a feature of Stokes process. In recent years, driven by the potential applications in bio-imaging, photovoltaics, solid-state lasers, optical anti-counterfeiting, optical information storage, and luminescence-based temperature sensing, the upconversion photoluminescence has emerged as a prominent research focus in the field of photonics [8–13]. Among these applications, temperature sensing based on the upconversion photoluminescence has gained considerable attention as an effective thermometry technique, offering distinct advantages such as non-invasive operation and immunity to electromagnetic interference [8]. The luminescence-based temperature sensing technique exploits temperature-dependent characteristics of emission

spectra, including fluorescence intensity ratios, normalized emission intensities, positions of emission peaks, and lifetimes [8–13].

To explore the mechanism of temperature sensing in the luminescent materials for enhancing the sensitivity, researchers developed various upconversion luminescent materials, including single crystals, polycrystalline ceramics and nanoparticles with different rare-earth ion dopants and host matrix. Liu et al. fabricated the Yb^{3+} , Er^{3+} co-doped $\text{Sr}_5(\text{PO}_4)_3\text{F}$ transparent ceramics and investigated the chromatically adjustable upconversion emission and luminescence thermometry based on the thermal coupled energy levels [14]. Ren et al. developed Yb^{3+} , Er^{3+} co-doped $\text{Gd}_3\text{Sc}_2\text{Al}_3\text{O}_{12}$ single crystal and achieve dual-mode luminescence thermometry [15]. Wei et al. reported the multiple luminescence intensity ratio thermometers based on both thermal coupled and non-thermal coupled energy levels of Ho^{3+} , Tm^{3+} co-activated $\text{Na}_3\text{Y}_9\text{F}_{32}$ single crystal [16]. Stamenković et al. optimized the quantum yield for upconversion photoluminescence and explored the luminescence intensity ratio based thermometry of Tm^{3+} activated SrGd_2O_4 nanoparticles via the emissions resulting from thermal coupled Stark sub-levels [17]. Among various trivalent lanthanide activators, Ho^{3+} ions are particularly appealing for upconversion applications due to the unique energy level structure, enabling strong green, red and infrared emissions in a wide wavelength range [18]. When co-doped with sensitizers such as Yb^{3+} ions, Ho^{3+} ions exhibit enhanced fluorescence intensity resulting from efficient energy transfer, making them highly suitable for luminescence thermometry [19].

The choice of host matrix significantly influences both sensitivity and operational range of luminescence thermometry [9,20]. Compared with the common fluoride matrices, oxide matrices have the advantages of non-toxicity and high thermal and chemical stability. $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$, a representative Aurivillius-type ferroelectric oxide, has unique layered crystal structure, composed of alternately stacked $[\text{Bi}_2\text{O}_2]^{2+}$ layers and perovskite blocks, produces a structurally anisotropic lattice with large A-site cations and octahedrally coordinated B-site cations, thereby providing a versatile crystal field environment for accommodating rare-earth dopants. This architecture facilitates the incorporation of sensitizer and activator ions, enabling efficient energy transfer and stable upconversion luminescence. Few studies have reported on its ferroelectric and piezoelectric properties achieved through doping, composite construction, and texture control [21,22]. However, little attention has been paid to its luminescent performance and luminescence-based temperature sensing when doped with rare-earth ions. This research gap provides an opportunity to explore the optical functionalities of $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ while fully leveraging the inherent advantages of oxide matrices in luminescence thermometry. Compared with conventional oxide hosts such as vanadates and molybdates, $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ offers several distinctive advantages, including a flexible layered framework that ensures structural stability even at relatively high doping concentrations. Additionally, the intrinsic anisotropy allows for controllable orientation (texturing), which is not feasible in most vanadates and molybdates hosts.

In this work, textured $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ ceramics with varying Ho^{3+} concentrations were fabricated through a combination of molten salt synthesis and solid-state sintering. Analysis of structural characteristics verified that Yb^{3+} and Ho^{3+} ions are readily incorporated into lattice, attributed to their ionic size and valence similarity to Bi^{3+} ions. The effects of Ho^{3+} content, excitation power and temperature on photoluminescence properties were systematically assessed. Furthermore, the temperature sensing properties of three characteristic upconversion emission bands under 980 nm laser excitation were evaluated using a thermal quenching model.

2. Materials and Methods

2.1. Materials Synthesis

Yb^{3+} , Ho^{3+} co-doped $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ powders with nominal composition of $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ($x = 0.01, 0.02, 0.03, 0.04$) were synthesized via the molten salt method. Stoichiometric amounts of raw materials, including CaCO_3 , Bi_2O_3 , TiO_2 , Yb_2O_3 and Ho_2O_3 were weighted according to the target compositions. NaCl and KCl were weighed in a 1:1 mass ratio and thoroughly mixed with the

reactants and an appropriate amount of anhydrous ethanol. The mixtures were ball milled at 300 r/min for 10 h in a nylon jar. The resulting slurry was dried at 120 °C in an oven. The dried mixtures were placed in a corundum crucible, and calcined at 850 °C for 5 h in a muffle furnace to achieve the molten salt reaction. The heating rate is 5 °C /min in the process of powder calcination. After calcination, the NaCl and KCl salts were removed by washing the resulting powders with deionized water preheated to 60 °C, followed by vacuum filtration. Filtration was performed using slow-flow quantitative filter paper with a pore size of 1–3 μm and the powders were thoroughly washed multiple times, gently resuspending between filtrations. After multiple rounds of washing and filtration, the insoluble products were collected and dried for the ceramics fabrication. The dried powders were mixed with a few drops of polyvinyl alcohol (PVA) binder and uniaxially pressed into disc-shaped pellets with diameter of 13 mm. For removing the binder, the green compacts were heated at 550 °C and held for 1 h in muffle furnace. The pellets were then sintered at 1200 °C for 2 h and cooled naturally to obtain the dense ceramics. The heating rate for sintering process is 5 °C /min.

2.2. Characterization

The phase structure and crystal orientation of the synthesized ceramic samples were examined by X-Ray diffraction (XRD). Upconversion photoluminescence (UCL) spectra of ceramics were excited using a power-adjustable 980 nm semiconductor laser and collected with a fibre-optic spectrometer. The spectral acquisition range was from 400 nm to 800 nm. To examine the excitation power dependence, the UCL spectra were recorded for each sample under a 980 nm laser excitation with different excitation powers. Additionally, the temperature-dependent UCL of ceramic samples with different compositions were investigated under a fixed excitation power of 200 mW. The samples were heated using a programmable heating stage, and UCL spectra were recorded at an interval of 20 K during the heating process, enabling systematic evaluation of optical response to temperature variations.

3. Results and Discussion

To investigate the structural characteristics of the synthesized $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ ceramics, XRD patterns were recorded. As shown in Figure 1 (a), the diffraction profiles of $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ($x = 0.01, 0.02, 0.03, 0.04$) ceramics exhibit the similar features, containing a series of sharp diffraction peaks with similar relative intensities and positions. These diffraction peaks can be indexed to (00L) reflections, including (006), (008), (0010), (0012), (0014), (0016), (0018) and (0020), demonstrating that the synthesized ceramics possess pronounced textural characteristics [23]. No additional phases associated with Yb^{3+} and Ho^{3+} doping ions were detected in the diffraction patterns, suggesting that the rare-earth ions were incorporated into host lattice within the experimental doping concentration range. Figure 1 (b) displays the enlarged view of the most intense (0010) diffraction peaks. As shown, a slight shift in peak position occurs with increasing Ho^{3+} concentration. With Ho^{3+} doping concentration increases, the (0010) diffraction peak initially shifts slightly toward lower angles, and with further doping, the peak then shifts toward higher angles. The observed subtle peak shift can be attributed to the substitution of the smaller rare earth ions for larger Bi^{3+} ions, which initially leads to a lattice expansion along the c-axis. With further doping, the continuous reduction in the average ionic radius of the perovskite layers results in lattice contraction.

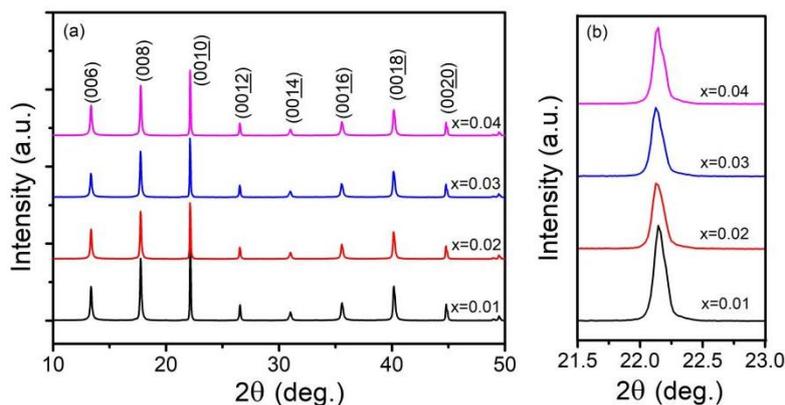


Figure 1. Structural characteristics of Yb³⁺, Ho³⁺ co-doped CaBi₄Ti₄O₁₅ textured ceramics (a) XRD patterns of Yb³⁺, Ho³⁺ co-doped CaBi₄Ti₄O₁₅ textured ceramics with varying Ho³⁺ concentrations; (b) enlarged view of the most intense (0010) diffraction peaks.

The interplanar spacing (d-spacing) of the (0010) plane can be calculated using following Bragg's law,

$$n\lambda = 2d\sin\theta \quad (1)$$

Based on the experimental XRD results of synthesized CaBi_{3.96-x}Yb_{0.04}Ho_xTi₄O₁₅ ceramic samples, the calculated d-spacings of the (0010) plane are 4.0105, 4.0136, 4.0141 and 4.0111 Å for x = 0.01, 0.02, 0.03 and 0.04, respectively. The evolution in d-spacing confirms the variation in Ho³⁺ ions doping concentration effectively induce a subtle modulation of the lattice structure.

For quantitatively evaluating the degree of preferred orientation in the CaBi_{3.96-x}Yb_{0.04}Ho_xTi₄O₁₅ ceramics, Lotgering factors were determined by the following formulas,

$$F = \frac{P - P_0}{1 - P_0} \quad (2)$$

$$P = \frac{\sum I(00L)}{\sum I(HKL)} \quad (3)$$

$$P_0 = \frac{\sum I_0(00L)}{\sum I_0(HKL)} \quad (4)$$

where *F* is the Lotgering factor, representing the degree of preferred orientation along (00*L*) direction. *P* represents the relative intensity ratio calculated from experimental XRD profile. *I*(00*L*) and *I*(*HKL*) are XRD peak intensities corresponding to the (00*L*) and (*HKL*) planes of the measured samples. *I*₀(00*L*) and *I*₀(*HKL*) are the peak intensities of (00*L*) and (*HKL*) planes from the standard PDF card No. 52-1640.

The values of *F* for synthesized CaBi_{3.96-x}Yb_{0.04}Ho_xTi₄O₁₅ ceramics with x = 0.01, 0.02, 0.03 and 0.04 are determined to be 0.918, 0.913, 0.916 and 0.917, respectively.

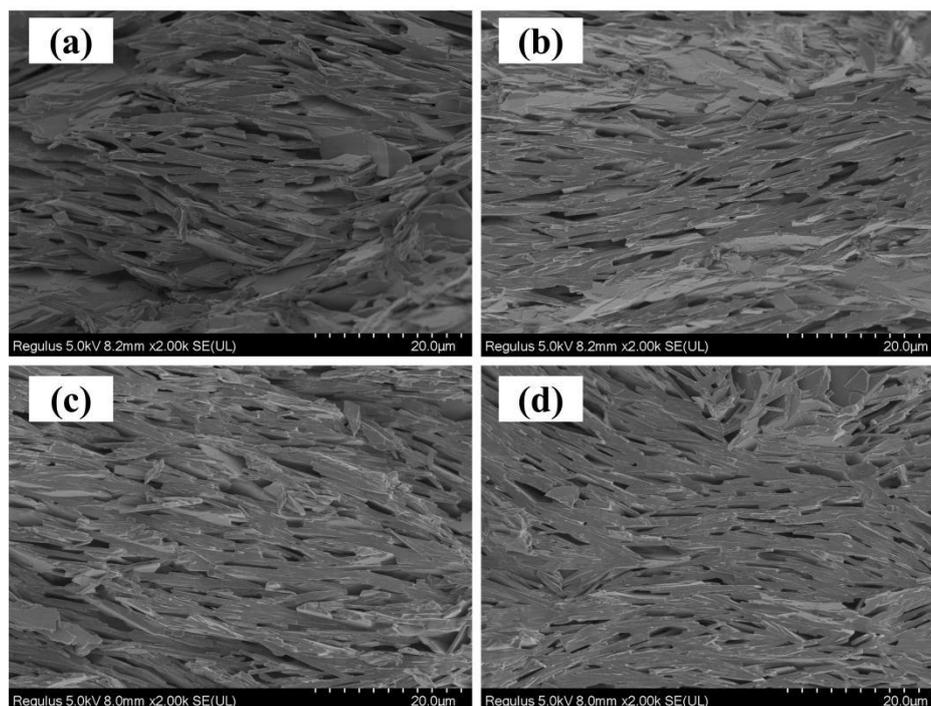


Figure 2. SEM images of the synthesized $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ceramic cross-sections corresponding to $x =$ (a) 0.01; (b) 0.02; (c) 0.03; (d) 0.04 (Scale bar: 20 μm).

The morphology of the Yb^{3+} , Ho^{3+} co-doped $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ ceramics were investigated using scanning electron microscopy (SEM). Figure 2 (a)–(d) present the SEM images of the $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ceramic cross-sections corresponding to $x = 0.01, 0.02, 0.03$ and 0.04 , respectively. As seen, the cross-sectional SEM images reveal the plate-like grains are aligned along the thickness direction, forming a well-defined lamellar stacking morphology, thereby indirectly confirming the pronounced textural orientation, which is consistent with the preceding XRD analysis.

Yb^{3+} ions were employed as the sensitizers at a fixed concentration, while the concentration of activator Ho^{3+} ions were varied. The upconversion emission spectra of textured ceramics with nominal composition of $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ($x = 0.01, 0.02, 0.03, 0.04$) were recorded in the wavelength range from 400 nm to 800 nm upon the excitation of continuous-wave 980 nm laser with fixed pump power at 200 mW, as shown in Figure 3 (a). The upconversion emission spectra revealed three distinct emission bands peaked at 549 nm, 660 nm, and 759 nm, corresponding to the transitions of $^5\text{F}_4 \rightarrow ^5\text{I}_8$ (Green emission), $^5\text{F}_5 \rightarrow ^5\text{I}_8$ (Red emission), and $^5\text{F}_5 \rightarrow ^5\text{I}_6$ (Infrared emission) of Ho^{3+} ions, respectively. The red emission band at 660 nm exhibits the highest emission intensity for all the four groups of textured ceramics. Figure 3 (b) shows the variation in the integrated intensities of three emission bands as a function of the concentration of Ho^{3+} ions. Within the investigated doping range, the emission intensity exhibits a monotonic increase with increasing dopant concentration. The result indicates the concentration quenching were not observed within the selected doping range, although this phenomenon may occur at higher doping concentrations [18].

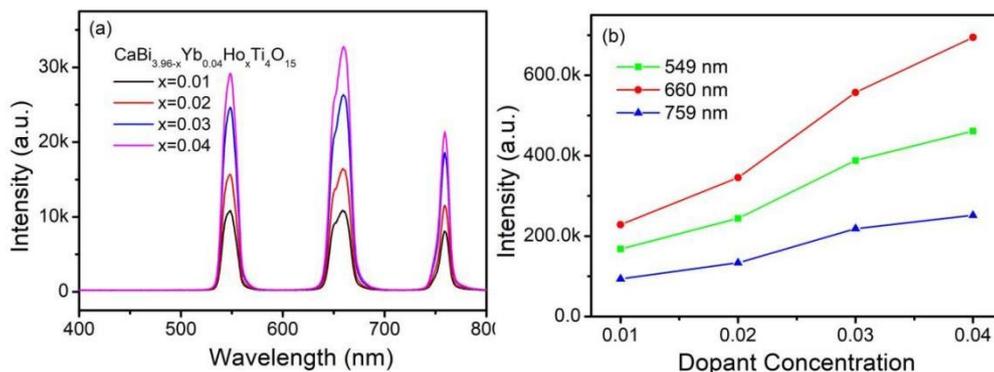


Figure 3. Upconversion emission spectra of Yb³⁺, Ho³⁺ co-doped CaBi₄Ti₄O₁₅ textured ceramics under 980 nm excitation (a) upconversion emission spectra of ceramics with different Ho³⁺ concentrations; (b) dopant concentration-dependent integrated intensities of three emission bands.

To further elucidate the mechanism of upconversion photoluminescence of the textured ceramics, the dependence of emission on excitation power was investigated. Figure 4 (a)–(d) present the upconversion emission spectra of different textured ceramics under identical testing conditions, with excitation power varying from 200 mW to 600 mW. It can be observed that the positions of the emission peaks remain unchanged with excitation power, while the emission intensity increases significantly. The dependence of emission intensity on excitation power can be described by the following relationship [24],

$$I \propto P^n \tag{5}$$

where I represents the emission intensity, P is the excitation power, and n is the number of photons involved in the excitation process [24].

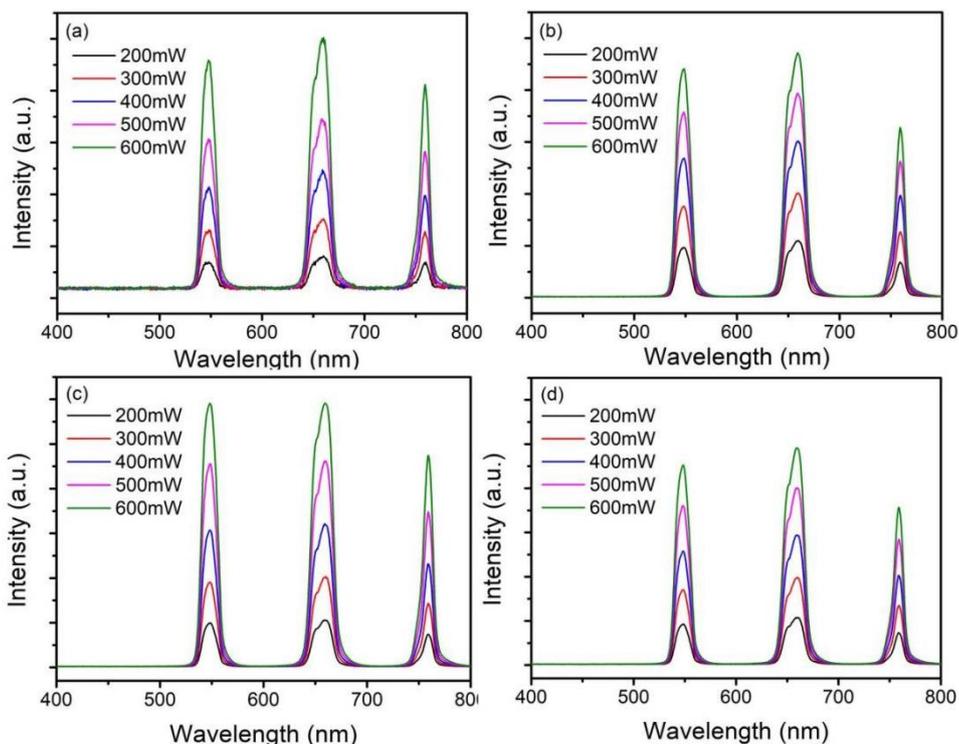


Figure 4. Power-dependent upconversion emission spectra of CaBi_{3.96-x}Yb_{0.04}Ho_xTi₄O₁₅ textured ceramics with x = (a) 0.01, (b) 0.02, (c) 0.03, (d) 0.04.

Based on the power-dependent emission spectra, the luminescence mechanism can be analyzed by plotting the dual-logarithmic coordinates of I and P. Figure 5 (a)–(d) present the experimental plots of $\log I$ versus $\log P$ along with the corresponding linear fitting results for green, red and near-infrared upconversion emissions of different textured ceramics. The slopes obtained from the linear fits represent the numbers of photons participating in the respective upconversion processes. For $\text{CaBi}_{3.95}\text{Yb}_{0.04}\text{Ho}_{0.01}\text{Ti}_4\text{O}_{15}$, the slopes for green, red, and near-infrared emissions are 2.03, 1.87 and 1.84, respectively. For $\text{CaBi}_{3.94}\text{Yb}_{0.04}\text{Ho}_{0.02}\text{Ti}_4\text{O}_{15}$, the corresponding slopes are 1.42, 1.37 and 1.48. In the case of $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$, the corresponding slopes are 1.70, 1.64 and 1.74, whereas for $\text{CaBi}_{3.92}\text{Yb}_{0.04}\text{Ho}_{0.04}\text{Ti}_4\text{O}_{15}$, the slopes are 1.49, 1.44 and 1.50. The slope values lie between 1 and 2 due to competing linear decay processes occurring during the upconversion process [25]. These results indicate that two-photon processes are responsible for the green, red and near-infrared emissions under 980 nm laser excitation in all investigated Yb^{3+} , Ho^{3+} co-doped $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ textured ceramics. At specific Ho^{3+} doping concentration, the significant decrease of the slopes can be explained by the in-band ${}^2\text{F}_{5/2}$ of Yb^{3+} becomes more efficiently depleted due to the enhanced $\text{Yb}^{3+} \rightarrow \text{Ho}^{3+}$ energy transfer. This leads to partial saturation of the first step of the two-step upconversion process under the applied excited power. The population growth rate of intermediate level in these phosphors can be reduced by the saturation.

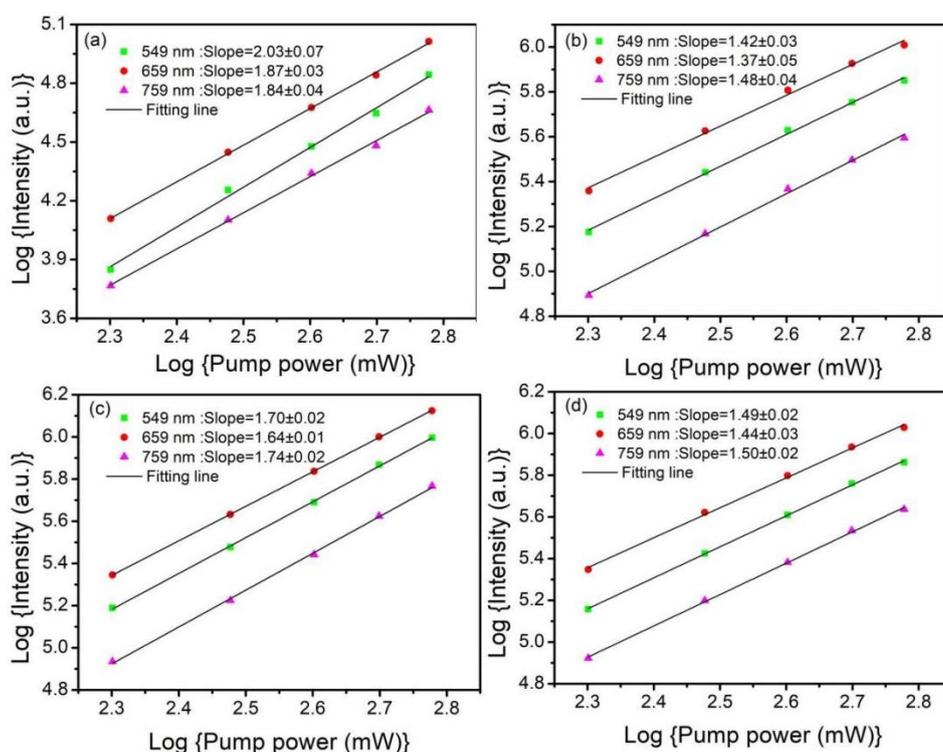


Figure 5. Dual-log plots of upconversion emission intensity versus excitation power at 549 nm, 660 nm and 759 nm for $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ textured ceramics with $x =$ (a) 0.01; (b) 0.02; (c) 0.03; (d) 0.04.

To shed light on the effect of temperature on upconversion photoluminescence of the synthesized textured ceramics, the temperature-dependent emission spectra of $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ($x = 0.01, 0.02, 0.03, 0.04$) were recorded upon excitation of 980 nm laser with a fixed excitation power of 200 mW. Figure 6 (a)–(d) present the emission spectra in range from 400 nm to 800 nm collected over the temperature range of 303 K to 503 K. It can be clearly observed that, for all four investigated compositions, the fluorescence intensities of three characteristic emission bands decrease monotonically with increasing temperature. By comparing the intensities of emission bands, it is evident that the $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ ceramic exhibits the largest variation in luminescence

intensity within the investigated temperature range, indicating the most significant fluorescence thermal quenching and highlighting its suitability for luminescence-based temperature sensing.

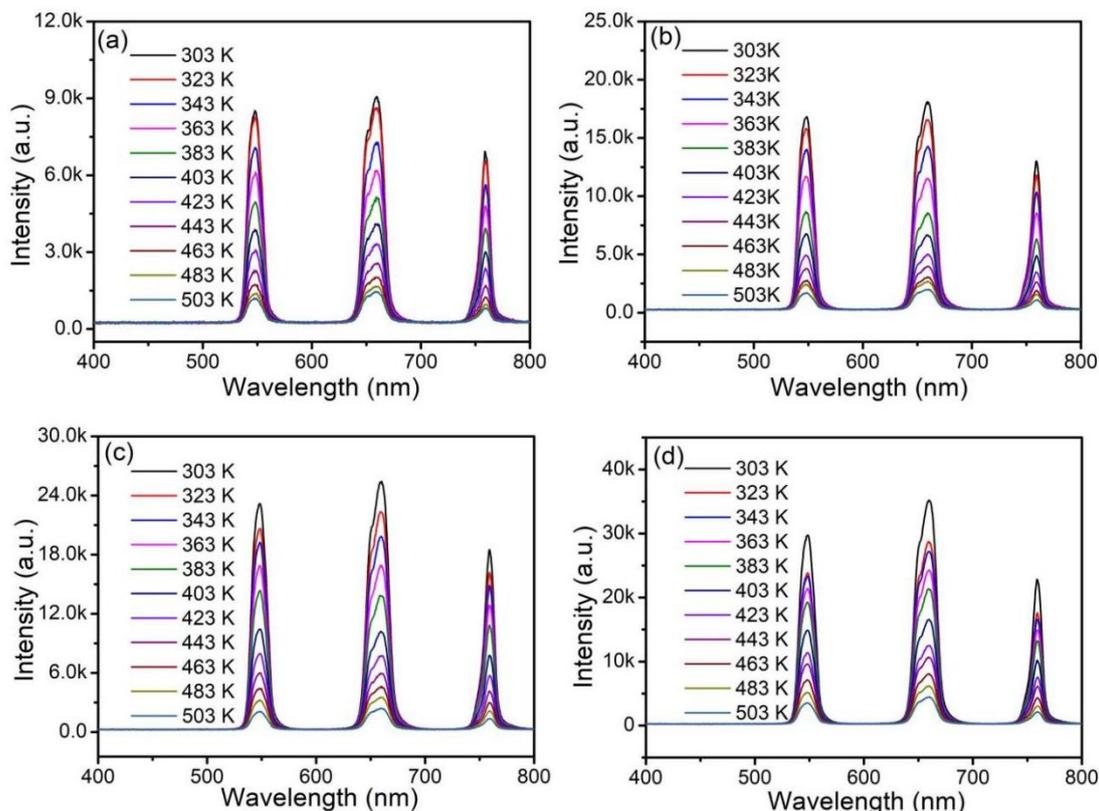


Figure 6. Temperature-dependent emission spectra of $\text{CaBi}_{3.96-x}\text{Yb}_{0.04}\text{Ho}_x\text{Ti}_4\text{O}_{15}$ ceramics under excitation of 980 nm laser for $x =$ (a) 0.01; (b) 0.02; (c) 0.03; (d) 0.04.

To quantitatively investigate the response of fluorescence intensity to temperature variation and gain deeper insight into the thermal quenching behavior of $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ textured ceramic, the relationship between temperature and normalized integrated intensities of the three distinct emission bands was analyzed. The basis for analysis of the observed temperature-dependent luminescence characteristics can be described via the following Arrhenius Equation (6) [26],

$$I(T) = \frac{I_0}{1 + C \exp\left(-\frac{\Delta E}{kT}\right)} \quad (6)$$

where $I(T)$ represents the integrated emission intensity at the T K, I_0 corresponds to the integrated emission intensity at 0 K, C represents the fitting coefficient, k is Boltzmann constant, ΔE denotes the activation energy associated with thermal quenching [26]. Figure 7 (a)–(c) illustrate the temperature-dependent normalized integrated intensities for the emission bands at 549 nm, 650 nm and 759 nm of $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ textured ceramic, along with the corresponding fitted curves. For normalization, the emission intensity for each band was divided by the value recorded at 303 K. According to the fitting analysis, the thermal quenching activation energies for 549 nm, 650 nm and 759 nm emission band were determined to be 0.36, 0.31 and 0.37 eV, respectively.

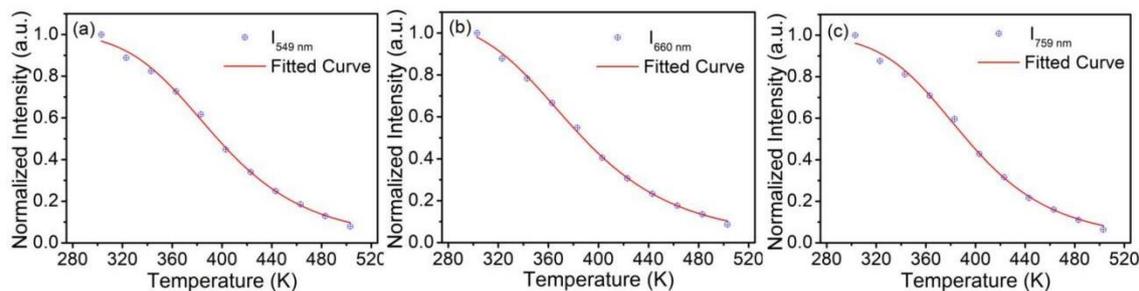


Figure 7. Normalized emission intensity as a function of temperature for $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ textured ceramic at (a) 549 nm; (b) 660 nm; (c) 759 nm emission.

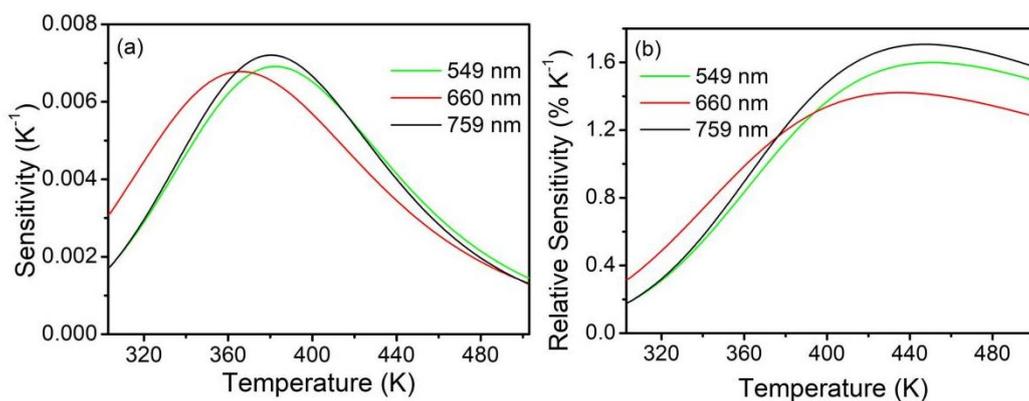


Figure 8. Absolute and relative sensitivity characteristics of $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ textured ceramic for optical thermometry based on fluorescence intensity (a) absolute sensitivity; and (b) relative sensitivity.

Sensitivity plays a critical role in evaluating the performance of synthesized luminescent textured ceramics in luminescence-based temperature sensing. Two sensitivity metrics, including absolute sensitivity (S_a) and relative sensitivity (S_r), are commonly employed for such assessments. The values of S_a and S_r can be defined by the following Equations (7) and (8), respectively [11].

$$S_a = \left| \frac{dI(T)}{dT} \right| \quad (7)$$

$$S_r = \frac{1}{I(T)} \left| \frac{dI(T)}{dT} \right| \times 100\% \quad (8)$$

As defined in Equations (3) and (4), S_a quantifies the change in fluorescence intensity per unit temperature variation, while the S_r expresses the change as a percentage of the intensity at the corresponding temperature. Figure 8 (a) presents the temperature-dependent S_a of luminescence thermometry based on three upconversion emission bands of $\text{CaBi}_{3.93}\text{Yb}_{0.04}\text{Ho}_{0.03}\text{Ti}_4\text{O}_{15}$ textured ceramic under 980 nm excitation. It can be observed that, for each upconversion emission bands, the S_a exhibits an initial ascent with increasing temperature, followed by a descent as the temperature continues to increase. The maximum values of S_a based on the 549 nm, 660 nm and 759 nm emission bands are 0.0069 K^{-1} , 0.0068 K^{-1} and 0.0072 K^{-1} , occurring at 382 K, 366 K and 380 K respectively. The S_r curves exhibit similar trends, increasing first and then declining with temperature, as shown in Figure 8 (b). The S_r for the 549 nm, 660 nm and 759 nm emission bands reach the maximum values of $1.60\% \text{ K}^{-1}$, $1.42\% \text{ K}^{-1}$ and $1.71\% \text{ K}^{-1}$, attained at 451 K, 436 K and 447 K, respectively. Among the three emission bands, the near-infrared emission at 759 nm demonstrates the best performance of luminescence thermometry within the investigated temperature range. These results indicate that Yb^{3+} , Ho^{3+} co-doped $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ textured ceramics hold promising potential for luminescence-based temperature sensing applications.

4. Conclusions

In conclusion, the Yb³⁺, Ho³⁺ codoped CaBi₄Ti₄O₁₅ ceramics were successfully synthesized via a molten salt method followed by solid state sintering. Based on both XRD analysis and microstructural observations, the synthesized ceramics exhibit pronounced textural characteristics. The upconversion luminescence of the synthesized textured ceramics under 980 nm excitation was confirmed to be governed by a two-photon process. Temperature-dependent spectra revealed significant monotonic thermal quenching in all three emission bands with the near-infrared emission at 759 nm demonstrating the optimal S_a of 0.0072 K⁻¹ at 380 K and S_r of 1.71% K⁻¹ at 447 K. These findings highlight the strong potential of the synthesized Yb³⁺, Ho³⁺ codoped CaBi₄Ti₄O₁₅ textured ceramics for luminescence-based temperature sensing.

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Availability of Data and Materials: The datasets generated and analyzed during the current study are available from the corresponding author upon reasonable request.

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Conflicts of Interest: The authors declare no conflict of interest.

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